Investigation of Constant Stack Pressure on Lithium-Ion Battery Performance

Aiden Leonard^a, Brady Planden^a, Katie Lukow^{a*}, Denise Morrey^a

^aHigh Voltage Energy Storage Group, School of Engineering, Computing, and Mathematics, Oxford Brookes University, Oxford, UK, OX33 1HX

*Correspondence: clukow@brookes.ac.uk

7 Abstract

1

2

3

4

5

6

Current research involving applying stack pressure to lithium-pouch cells has shown both 8 performance and lifetime benefits. Fixtures are used to mimic this at the cell level and 9 conventionally prescribe a constant displacement onto the cell. This increases stack pres-10 sure, but also causes pressure to vary. Despite this, applying an initial stack pressure 11 improves cell conductivity and cell lifetime [1, 2]. In this work, a fixture was designed 12 that applies constant pressure to the cell independent of displacement. The fixture uses 13 pneumatics to apply a constant stack pressure independent of elastic and plastic swelling. 14 Cells constrained by the constant pressure fixture and a conventional displacement based 15 fixture were evaluated using a Hybrid Pulse Power Characterisation (HPPC) test to mea-16 sure internal resistance and maximum deliverable power. Multiple stack pressures were 17 applied to investigate the variance in pressure over operational conditions and perfor-18 mance between constant pressure and constant displacement based methods. All tests 19 were further compared to a control case with no applied stack pressure. The constant 20 pressure based method reduced pressure variation during charging and discharging, re-21 duced the discharge impedance and improved discharged power, but did not improve 22 charge performance. Discharge performance benefits from constant pressure could influ-23 ence pack design to improve vehicle performance. 24

²⁵ Keywords: Lithium-ion battery, Pack design, Stack pressure, Battery performance

²⁶ Abbreviations

Symbol	Definition				
Q_{Δ}	Capacity loss/gain (Ah)				
CPF	Constant pressure fixture				
DCIR	Direct current internal resistance				
D_{\max}	100% Maximum discharge current				
$D_{\max/2}$	50% Maximum discharge current				
HPPC	Hybrid pulse power characterisation				
MBPF	Modular battery pressure fixture				

27 1 Introduction

Lithium-ion cells have quickly become to standard for many industries requiring reliable 28 and efficient battery storage. Pouch cells provide a unique solution for increased packag-29 ing density and increased power density when compared to most conventional cylindrical 30 cells; however, they bring additional challenges as well. Most notably, is the requirement 31 of external stack pressure to prolong life and optimise performance. Stack pressure has 32 been applied to pouch cells via various methods, generally falling into two categories, 33 fixed displacement and constant pressure. Conventionally, fixed displacement is achieved 34 by constraining the orthonormal expansion of the cell through rigid plates. Constant 35 pressure based methods conventionally allow for expansions of the cell through the ad-36 ditions of varying-stiffness foam or spring elements [1, 2]. Pressure has been shown to 37 improve the interfacial surface area between the negative electrode, positive electrode, 38 and separator, thus decreasing the ionic resistivity [3–6]; however, reaches a critical value 39 where additional mechanical stress has been shown to reduce active electrode material, 40 reducing the performance of cells [5–8]. Stack pressure varies elastically throughout the 41 battery's state-of-charge for a corresponding fixed displacement fixture due to lithiation 42 of the anode and increases over time due to anode growth [1, 9–12]. Development of a 43 stack pressure method that is cell thickness agnostic is the aim of this work, potentially 44 providing performance benefits through increasing the positive effects of pressure without 45 causing damage through uncontrolled pressure increases due to ageing [1]. 46

Current research involving applying stack pressure to pouch cells has resulted in im-47 mediate and long-term performance benefits. A study conducted by Müller et al. [5] 48 utilised parallel plates with springs to apply pressure ranging from 0-0.84 MPa to both 49 a full NMC/graphite cell and the individual cathode, anode, and separator. The results 50 show an optimal pressure to minimise separator resistivity from 0.1-0.6 MPa, and an 51 increasing relationship between the electrode resistances and pressure. At the cell level, 52 stack pressure increased the charge transfer resistance but decreased the high frequency 53 resistance. Pressures above 100 kPa have been seen to improve conductivity for future 54 cell materials, such as lithium-metal and solid electrolytes [13–15]. Doux et al. [13] ex-55 plored the effect of stack pressure on a sulfide electrolyte solid-state battery and tested 56 pressures from 5 MPa to 70 MPa. Electrode conductivity improved for pressures up to 57 70 MPa, while discharge capacity decreased at the upper limit of pressure tested. A 58 study conducted by Louli et al. [16] found that 1.7 MPa of stack pressure provided the 59 highest performance for a lithium-metal negative electrode cell using a liquid electrolyte; 60 However, the study reported a 50-300% change in pressure from the thickness change of 61 the cell during charging and discharging. A hybrid lithium-ion/lithium-metal cell was 62 also found to benefit from 1.2 MPa of applied stack pressure, [17] enabling a dendrite 63 suppression mechanism which corresponds to cycle-life benefits. For lithium-ion cells, 64 the SEI layer has been shown to grow over the life of the cell, increasing impedance and 65 decreasing usable capacity [18]. Stack pressure is shown to reduce capacity fade through 66 suppressing delamination of electrodes, gassing of the electrolyte, and SEI layer growth 67

[7, 11]. Hahn et al. [1] presents a varying applied stack pressure between 38-580 kPa, 68 improved capacity retention from 95% to 99% after 70 days of calendar ageing. Further 69 studies support the discharge capacity improvement gained from reducing the applied 70 current density due to pressure application [19, 20]. Along with capacity improvements, 71 increasing stack pressure for lithium-ion cells has shown to improve interfacial contact 72 of electrodes to the separator [7]. Since non-flat electrode surfaces have a limited con-73 tact surface area, creating a more ideal flat surface contact between elements in cells 74 results in immediate performance benefits. With elastic contact on rough surfaces, the 75 contact area increases proportionally to the load [21–23]. Improving interfacial surface 76 area contact immediately reduces the current density in the localised region [20]. The 77 larger interaction area between electrodes also reduces the effective ion path length, fur-78 ther reducing impedance [15]. These performance benefits from applying a stack pressure 79 influence current and next-generation battery pack design. Current modules have two 80 main methods of applying stack pressure. Modules fix the outer dimensions of cells using 81 cylindrical cells or volumetrically constricted groups of pouch cells [24]. Furthermore, 82 deformable materials are used between cells to reduce pressure variance from expansion 83 and contraction [1]. Based on current research on lithium-metal [14, 17] and Silicon [13, 84 19, 25] cells, future battery packs will likely benefit from higher stack pressure applied 85 to cells. Studies look for performance benefits by either constraining thickness or using 86 spring-like elements. 87

Basic fixtures use flat parallel plates and apply pressure by using bolt torques to 88 clamp the cell between the plates [13, 26, 27]. However, because the width between 89 each plate is essentially fixed, stack pressure varies during charging and discharging due 90 to elastic swelling, with SOC due to differences in electrode volumes, and over time 91 increases due to negative electrode growth [1, 28–30]. Hahn et al. [1] studied the long-term 92 effects of mechanical pressure by using a hydraulic cylinder and porous foam as a spring 93 element. This approach provided flexibility in altering pressure to model cell elasticity 94 as a spring-like element; however, this study did not observe the effects of constant 95 pressure due to the pressure increase over cell lifetime corresponding with cell thickness 96 growth. Other novel fixtures [2, 17], utilise buffer layers of foam to dampen the thickness 97 growth; however, stack pressure then becomes dependent on the compressive stiffness of 98 the foam. Conventionally, to apply a constant, high amplitude pressure, three methods 99 are utilised: electric, hydraulic, or pneumatic actuation. Using pneumatic actuation has 100 conventionally provided advantages of low viscosity and compressibility, thus minimising 101 the pressure variance to a corresponding volume change; however, a system leakage is 102 common causing the need for an air compressor. Hydraulic actuation commonly provides 103 the lowest cost with minimal leakage under normal operation; however, even minimal 104 hydraulic leaks could cause an electrical short circuit for the tested battery. Electric 105 actuation can provide a constant pressure over long periods, but the corresponding high 106 power consumption and pressure dependency on motor and sensor accuracy are not ideal. 107 Due to the above limitations, hydraulic and electric actuation were ruled out due to the 108 risk of short circuits and high costs. As pneumatic actuation does not suffer from these 109

¹¹⁰ limitations and has a relatively low cost, it was selected for this work.

The performance impacts of constant pressure on lithium-ion pouch cell is relatively unknown. As previously discussed, constant pressure research has been previously focused on low amplitude (<40 N Jiang et al. [2]) or amplitudes above 1 MPa for lithium-metal chemistries [14]. In this paper, a constant pressure fixture (CPF) utilising pneumatic actuation for stable pressure values independent of elastic and plastic swelling is presented.

116 2 Methodology

The following section provides an overview of the fixture design, data acquisition and analysis methods, and experimental methods.

¹¹⁹ 2.1 Fixture Design

A novel fixture was designed to maintain a constant face pressure during cell cycling using 120 a pneumatic actuator. The design targeted up to 180 kPa for testing current-generation 121 liquid electrolyte cells with the ability to replace the pneumatic actuator to allow for 122 larger face pressures if required. Figure 1 presents the design of the proposed constant 123 pressure fixture (CPF) and the reference constant displacement fixture, referred to as the 124 modular battery pressure fixture (MBPF). The fixture applies a constant stack pressure 125 to the face of the battery through the pneumatic actuator and is transferred through two 126 carbon-inlaid 3D-printed plates. This material electrically isolates the battery to prevent 127 the risk of short circuits and provides sufficient stiffness to improve pressure distribution. 128 The ball-and-socket joint provides rotational freedom, allowing the contact between the 129 cell and the pressure plates to be uniform and less dependent cell swelling. Two TE FX29 130 load cells measure force, that are monitored through a Teensy 4.1 board and recorded 131 onto a microSD card. 132



Figure 1: CPF (top) and MBPF [31] (bottom) CAD, with crucial design elements enumerated.

Results in this work were compared against two other fixture methods. A baseline 133 condition of no external stack pressure was first tested. Second, a constant displacement 134 fixture developed by the High Voltage and Energy Storage group as shown in figure 1 135 [31]. The fixture applies stack pressure through two plates fastened at up to 6 locations, 136 measured through TE FX29 sensors similar to the constant pressure fixture. Further 137 information can be found in the GitHub repository. As discussed, stack pressure was 138 applied through a pneumatic piston connected to an air reservoir to counteract cell swell 139 and minor leaks within the system. Initial testing showed that pressure was maintained 140 over a 48 hour period. 141



Figure 2: Test measuring pressure variation over 24 hours between the MBPF (left) and CPF (right).

Two TE FX29 load cells were placed between the lower cell plate and the base of the test fixture. The load cells were connected to a Teensy 4.1 microcontroller that recorded the values throughout the test via a microSD card. A type-T thermocouple was placed on the body of the cell located near the cell tabs. An Arbin LBT-21084-HC cell cycler was used to perform the experiments.

$_{147}$ 2.2 Test method

A 3.7 Ah LCO/graphite pouch cell was used throughout this study with specifications as
 defined in Table 1.

Cell	Chemistry	Nominal Voltage [V]	Initial AC	Initial DC	Nominal	Energy	Power
			Impedance	Resistance	Capacity	Density	Density
			$m\Omega$	$m\Omega$	[Ah]	[Wh/kg]	[W/kg]
Melasta	LCO / NMC	3.8	< 2.6	N/A	3.7	204	5043
SLPB							
$7336128\mathrm{HV}$							

Table 1: Rated Cell Specifications [32]

A Hybrid Pulse Power Characterisation (HPPC) test was conducted every 5% stateof-charge, beginning at 100% SOC. A pulse profile a 10 second load followed by a 40 second rest was completed as shown in Figure 3.



Figure 3: HPPC Pulse Profile

The test was performed at the maximum discharge and maximum charge as shown in the figure above. Tests were also completed at half these values. Stack pressures were compared at 30, 60, and 90 kPa alongside a benchmark test that had no stack pressure applied. Ambient temperature was fixed at 25°C for all conditions.

157 **3** Results and Discussion

¹⁵⁸ 3.1 Pressure Variance

Pressure data was recorded for all 21 experiments. For all experiments, pressure increased 159 respective to both SOC and pulse current. Pressure varied more with the MBPF over 160 the tests, for 60 kPa of initial stack pressure, the MBPF pressure varied from 44-171 161 kPa, while the CPF cell pressure varied from 54-69 kPa. The measured stack pressure 162 increased during both the charge and discharge current pulses (Figure 4). The relationship 163 between pressure and SOC for each pulse (Figure 5) shows The CPF having a linear slope 164 with an increased slope above 60% SOC; however, the MBPF's fixed displacement method 165 resulted in a large pressure vs SOC slope compared to the CPF. While the MBPF provides 166 poor performance across the full SOC operational range, within 30-60% it has a small 167 range of potential acceptable usage with a delta of 26.7-56.7%. The MBPF pressure vs 168 SOC slope was lower for 90 kPa of initial stack pressure at above 80% SOC, compared to 169 30 kPa and 60 kPa. This could be due to physical deformation of the cell orthogonal to 170 the clamping force, or due to deformation of the MBPF itself. According to Hahn et al. 171 [1] and Li et al. [6], increased cell deformation occurs above 1000 kPa, therefore the most 172 likely cause of the decline in slope of pressure vs SOC is the elastic creep of the MBPF 173 fixture itself. For the MPBF, significant changes in pressure occur at approximately 30% 174 and 60% SOC. This is expected to correspond with the knee points in the open-circuit 175 potential as per Figure 4, as the thickness of the cell aligns with the voltage vs SOC curve 176

177 [28].



Figure 4: The pressure, voltage, and current throughout the test for CPF (top) and MBPF (bottom) at 60 kPa of stack pressure.



Figure 5: The mean stack pressure for the 10-second discharge pulse for CPF (left) and MBPF (right) for three initial pressures across state-of-charge.

The CPF provides a reduction in pressure variance and as such improves future 178 pouch cell related pressure independence studies. For example, he MBPF stack pressure 179 increased up to 317% of the initial value for 30 kPa, while the CPF increased by 6%. By 180 utilising the CPF, variance in pressure has been shown to be within +/-25%, reducing 181 pressure variance disruption on results. Since stack pressure has been shown to affect 182 discharge capacity over cycle life, [3, 5, 12, 17], improved pressure control would enable 183 pressure invariant isolation of these effects. For example, excessive stack pressures can 184 lead to crack development in the electrode active material, with the CPF's ability to 185 adapt to varying thickness this mitigates this mechanism and further provides clarity on 186 the cell lifetime for a given pressure. 187

Transient pressure variations can occur due to the heat generation occurring inside the cell. Cells produce heat primarily from joule heating, introduced as,

190

$$Q = I^2 R \tag{1}$$

191

where *I* is the current through the cell and *R* is the internal resistance of the cell [33, 34]. As current was applied during the pulses, the cell temperature correspondingly increased. This results in cell swelling [35, 36] and therefore pressure should the pouch cell have its displacement constrained. The pressure variance during pulses (Figure 4) was similar between the MBPF and CPF, although the MBPF did have a higher variance. A reason the CPF may have performed similarly to the MBPF could be its reduced ability to adjust to cell thickness changes in short time frames. Friction between moving and static components may prevent the CPF from adjusting quickly enough to displacement changes to keep stack pressure constant in more transient scenarios. In the case of a battery pack, logging stack pressure to measure transient changes could be useful to gain information on cell energy and heat generation, in addition to temperature management.

Additionally, lithium-ion cell thickness growth over time due to SEI layer growth and 203 reduced packing efficiency further emphasises the importance of the CPF for degradation 204 testing. As the cell thickness increases during ageing, a constant displacement constraint 205 would result in rising pressures over time. This could lead to mechanical damage, chem-206 ical degradation, and premature failure due to excessively increasing stack pressures [7, 207 26. Using a constant pressure constraint would keep pressures more level even as the cell 208 degrades. This would allow for a more accurate degradation analysis for a given pressure. 209 The CPF could provide the capability of conducting degradation testing at various pres-210 sures with accurate SOH and failure results. A cycling ageing experiment using the same 211 pressure values and fixtures with a 1C standard charge and discharge could be conducted 212 to compare capacity loss between constant displacement and constant pressure. Fol-213 lowing the experiment with a postmortem scanning electron microscope, analysis could 214 reveal any physical and chemical degradation effects on cells from the pressure application 215 method. 216

217 **3.2** Cell Performance

Throughout this study, DC internal resistance was measured through the HPPC pulse and is defined as,

220

$$R = \frac{V_f - V_0}{I} \tag{2}$$

221

where V_f is the voltage measured at the end of the 10-second pulse, V_0 is the voltage at 222 the beginning of the pulse and I is the average current applied over the 10-second pulse. 223 A clear difference emerged in both charge and discharge DCIR between the CPF and 224 MBPF while initial pressure varied results for both the CPF and MBPF DCIR. For all 225 initial pressures, the CPF condition generally outperformed the MBPF for both discharge 226 and charge DCIR. Both the CPF and MBPF had the lowest discharge DCIR values at 227 30 and 60 kPa, while the benefits decreased at 90 kPa. The change in DCIR measured 228 by the CPF and MBPF compared to the control condition with 0 kPa of stack pressure 229 can be seen in Figures 6 and 7. The CPF and MBPF results are plotted against each 230 other at each initial pressure for both the D_{max} cycle and the $D_{\text{max}/2}$ cycle. These plots 231 show the difference in DCIR of the CPF and MBPF compared to the control condition, 232 indicated by the dashed line at y = 0. 233



Figure 6: Percent change in discharge DCIR vs SOC for the CPF and MBPF from the control condition at various initial pressures.

Note for Figures 6 and 7: The unconstrained control condition applies 0 kPa of pressure to the cell. The difference in DCIR between each of the two fixtures and the control condition are plotted at each SOC, with the control condition indicated with the dashed line.



Figure 7: Percent change in charge DCIR vs SOC for the CPF and MBPF from the control condition at various initial pressures.

For the discharge pulses (Figure 6), the CPF had lower DCIR than both the MBPF 234 and control conditions for SOCs below 80%. Above 80% SOC, the CPF only had lower 235 DCIR at 90 kPa initial pressure. The MBPF generally had lower DCIR than the control 236 condition in discharge for $D_{\text{max}/2}$, except for 90 kPa. For D_{max} , the MBPF discharge 237 DCIR was unanimously higher than the control condition. The CPF stands out as having 238 lower discharge DCIR than both the MBPF and the 0 kPa condition for all pressures 239 and both D_{max} and $D_{\text{max}/2}$. Holding pressure at a level value seemed to reduce discharge 240 internal resistances, especially at SOCs below 70%. This coincides with the pulse pressure, 241 as the CPF has a steeper pressure increase at SOCs above 70%. These benefits could come 242 from effectively increasing surface area through pressure application, without excessively 243 pressurising the cell. At Low SOCs, both the MBPF and CPF had the largest decreases 244 in internal resistance compared to the 0 kPa test, indicating that applied stack pressure 245 may have extra benefits at low SOCs. However, low SOCs are the point of the highest 246

²⁴⁷ DCIR so the normalised difference in DCIR would indicate that the reduction in DCIR ²⁴⁸ is proportional to the nominal value. The lesser difference in discharge DCIR above 70% ²⁴⁹ SOC may be because DCIR is less dependent on pressure at high SOC. Both fixtures ²⁵⁰ had fewer improvements in DCIR from the control condition at D_{max} . Between 30 and ²⁵¹ 60 kPa seemed optimal for both fixtures in terms of discharge resistance. 90 kPa may ²⁵² be excessively high for the MBPF, as the peak pressure reaching nearly 200 kPa could ²⁵³ mitigate the benefits of pressure.

For the charge pulses (Figure 7), the CPF generally had lower DCIR than the control 254 condition for $D_{\max/2}$, except for high SOCs where it had higher internal resistances. Both 255 the CPF and MBPF had higher charge DCIR than the control condition for D_{max} . The 256 MBPF had higher charge internal resistances at lower SOCs than the control condition 257 but had similar charge internal resistances at higher SOCs. The CPF had a lower charge 258 DCIR than the MBPF for nearly all cases, except high SOCs for $D_{\text{max}/2}$. Applied stack 259 pressure could reduce charge performance, which is worse at higher C-rates. Similar 260 to discharge DCIR, the 30 kPa and 60 kPa conditions seem more optimal than 90 kPa 261 stack pressure. Pressure may negatively affect charge resistance due to the decrease in 262 thickness with SOC due to the anode volume change. The applied pressure could be a 263 driving force that biases discharge, as discharging the cell over time decreases thickness. 264 For a 10-second pulse conducted in this study, it is difficult to evaluate if this effect 265 explains the difference in discharge and charge resistance compared to having no stack 266 pressure. The CPF at half the maximum current was the only beneficial condition for 267 charge DCIR. Further investigation into this effect could reveal nuances of the effect of 268 pressure on charge DCIR. 269

The maximum current D_{max} trial resulted in a lower charge and discharge DCIR for 270 both fixtures and all pressures, including the control. The lower DCIR for the D_{max} cur-271 rent cycle could be due to the higher prescribed current changing the plating mechanisms 272 of the electrode [37]. Higher current can accelerate electrochemical processes such as 273 the double layer discharging quicker, reducing the DCIR [38]. This poses an interesting 274 idea that higher current demands could reduce heat generation for pulse conditions in 275 performance settings. This could explain the benefits of pulse charging at certain cur-276 rents, where resistance is lower than steady charging, improving charging efficiency and 277 fast charging times. The temperature was higher for the D_{max} condition because higher 278 battery power results in higher heat generation. Since temperature only varied by 1°C, 279 it most likely did not affect the DCIR [37]. Both discharge and charge DCIR had maxi-280 mum values at the lowest SOC point for all trials. Discharge DCIR values were generally 281 lowest within the 30% to 60% SOC range, while charge DCIR values have a similar dip 282 in the 30% to 60% range, with their lowest value near 100% SOC. DCIR increased at 283 low SOC due to the reduction in available intercalation space in the cathode. Diffusion 284 becomes more difficult as more lithium ions occupy available space in the cathode ma-285 terial, increasing resistance. Inversely, the charge DCIR increased at high SOC, due to 286 the increased difficulty of intercalating lithium into the negative electrode. The charge 287 DCIR had less of a resistance increase, which aligns with previous studies [39–41]. 288

Power differences were also measured between the fixtures. Figures 8 and 9 show the 289 power plotted as a difference in discharge and charge power at various pressures compared 290 to the control baseline, shown by the dashed line at y = 0. Generally, both the discharge 291 and charge power increased with SOC, but the charge power was lowest at 95% SOC. 292 Power increased with SOC due to the cell voltage vs SOC. Discharge power at low SOC 293 and charge power at high SOC were both important metrics because minimum voltage 294 and maximum voltage limit the power, respectively. At high SOCs, being able to keep cell 295 voltage below the maximum cutoff voltage enables faster charging, while at low SOCs, 296 maintaining a voltage above the minimum cut-off voltage enables higher discharge power. 297 The CPF had higher discharge power than both the MBPF and control case for nearly 298 all pressures and SOCs, except for 60 kPa of stack pressure. Increasing discharge current 299 increased the difference in discharge power between the CPF and MBPF to the control 300 condition. The CPF had greater power benefits at the higher current, while the MBPF 301 had greater power detriments. The greater difference between the CPF and MBPF 302 at D_{\max} reveals that constant pressure could be more beneficial in terms of discharge 303 power at high C-rates. The MBPF performed worse at higher C-rates, indicating that 304 constraining displacement can be detrimental to cell performance in this scenario. The 305 CPF had the largest increase of power at low SOCs, except for the 90 kPa condition. The 306 CPF achieved a power difference on the last discharge pulse of over 3 W compared to 0307 kPa and 5 W compared to the MBPF when both fixtures were tested at 60 kPa. The 308 CPF saw this smallest increase of power at 90 kPa, possibly due to pressure exceeding 309 the limit of benefit for the cell. Similarly to DCIR, differences in charge power were less 310 significant between the fixtures than discharge power. Both the CPF and MBPF had less 311 charge power at high SOCs than the control condition, and slightly more charge power 312 at low SOC. The loss of charge power at high SOC could be because of the previously 313 mentioned idea that pressure can be adverse for charging in some cases. The MBPF had 314 an edge over the CPF for charge power, especially at low SOCs for D_{max} . The CPF had 315 less charge power than the control case for low SOCs at $D_{\rm max}$, performing worst at 90 316 kPa. 317



Figure 8: Percent change in discharge power vs SOC for the CPF and MBPF from the control condition at various initial pressures.

Note for Figures 8 and 9: The unconstrained control condition applies 0 kPa of pressure to the cell. The difference in power between each of the two fixtures and the control condition are plotted at each SOC, with the control condition indicated with the dashed line.



Figure 9: Percent change in charge power vs SOC for the CPF and MBPF from the control condition at various initial pressures.

Discharge capacity ranged from 3.84–3.86 Ah, for all fixtures constraints and for the 318 control. Given that the differences in discharge capacity were less than 1%, there is 319 not enough evidence to show that stack pressure affected discharge capacity in the short 320 term. Lithium-ion pouch cells may not benefit from the capacity increase from stack 321 pressure as with lithium-metal anode and silicon-blend anode cells, where much higher 322 stack pressures showed improvements in capacity [19, 26]. Hahn et al. [1] found that 323 stack pressure decreased lithium-ion cell capacity initially, then provided better capacity 324 retention during calendar ageing. The possible benefits of dendrite growth suppression, 325 gas suppression, and SEI layer growth suppression would only emerge with degradation 326 testing and/or calendar ageing. 327

328 4 Conclusion

A fixture was developed to evaluate the effects of constant pressure and constant displace-329 ment constraints on cell performance. The designed fixture performed as expected with 330 pressure variations of below 25% when compared to a conventional fixed-displacement 331 system with a pressure variation of over 300%. Improvements in discharge resistance 332 and power were observed by applying constant pressure with no significant capacity or 333 Coulombic efficiency differences were measured. Incorporating more uniform pressure on 334 pouch cells independent of cell swelling could improve discharge capabilities for perfor-335 mance scenarios. Designing battery packs that pressurize pouch cells while allowing them 336 to expand and contract could improve the discharge power of packs, an important metric 337 for performance scenarios. Additionally, lower discharge internal resistance would reduce 338 power loss during discharge, improving vehicle performance. 339

Further work could reduce errors from sensors and mechanical flex to obtain higher 340 The load cells measuring the pressure did have signal noise, although fidelity data. 341 this was seen to be less than the change in pressures during the discharge and charge 342 pulses. Nevertheless, hysteresis error and random error could have affected the pressure 343 results. Incorporating a singular, more accurate load cell could improve the resolution 344 and accuracy of the pressure data. Flexing in the pressure plates was seen during testing 345 for both the CPF and MBPF, and was more noticeable at 90 kPa. This deformation 346 could have negatively impacted pressure distribution, reducing the possible benefits of 347 stack pressure. Selecting a different design for the plates in terms of materials or geometry 348 could mitigate this possible source of error. Testing cell degradation with both fixtures 340 could reveal possible long-term capacity benefits from applying constant stack pressure. 350

351 References

- Severin Hahn et al. "Pressure Prediction Modeling and Validation for Lithium-Ion Pouch Cells in Buffered Module Assemblies". In: *Journal of Energy Storage* 40 (2021), p. 102517. ISSN: 2352-152X. DOI: https://doi.org/10.1016/j.est.
 2021.102517. URL: https://www.sciencedirect.com/science/article/pii/ S2352152X21002656.
- Yihui Jiang et al. "A stack pressure based equivalent mechanical model of lithium ion pouch batteries". In: *Energy* 221 (2021), p. 119804. ISSN: 0360-5442. DOI:
 https://doi.org/10.1016/j.energy.2021.119804. URL: https://www.
 sciencedirect.com/science/article/pii/S0360544221000530.
- [3] Abdilbari Shifa Mussa et al. "Effects of external pressure on the performance and ageing of single-layer lithium-ion pouch cells". en. In: *Journal of Power Sources* 385
 (May 2018), pp. 18–26. ISSN: 03787753. DOI: 10.1016/j.jpowsour.2018.03.020.
 URL: https://linkinghub.elsevier.com/retrieve/pii/S0378775318302441
 (visited on 07/18/2021).
- [4] Long Zhou et al. "A study of external surface pressure effects on the properties for
 lithium-ion pouch cells". In: International Journal of Energy Research 44.8 (2020),
 pp. 6778–6791.
- ³⁶⁹ [5] Verena Müller et al. "Study of the influence of mechanical pressure on the performance and aging of Lithium-ion battery cells". In: *Journal of Power Sources* 440 (2019), p. 227148. DOI: https://doi.org/10.1016/j.jpowsour.2019.227148.
- Ruihe Li et al. "Effect of external pressure and internal stress on battery performance and lifespan". In: *Energy Storage Materials* 52 (2022), pp. 395-429. ISSN:
 2405-8297. DOI: https://doi.org/10.1016/j.ensm.2022.07.034. URL: https:
 //www.sciencedirect.com/science/article/pii/S2405829722004044.
- John Cannarella and Craig B. Arnold. "Stress evolution and capacity fade in constrained lithium-ion pouch cells". In: *Journal of Power Sources* 245 (2014), pp. 745– 751. ISSN: 0378-7753. DOI: https://doi.org/10.1016/j.jpowsour.2013.
 06.165. URL: https://www.sciencedirect.com/science/article/pii/ S037877531301197X.
- [8] Christina Peabody and Craig B. Arnold. "The role of mechanically induced separator creep in lithium-ion battery capacity fade". In: Journal of Power Sources
 196.19 (2011), pp. 8147–8153. ISSN: 0378-7753. DOI: https://doi.org/10.1016/ j.jpowsour.2011.05.023. URL: https://www.sciencedirect.com/science/ article/pii/S037877531100989X.

- Shaojun Niu et al. "Analysis on the effect of external press force on the performance of LiNi0.8Co0.1Mn0.1O2/Graphite large pouch cells". In: *Journal of Energy Storage* 44 (2021), p. 103425. ISSN: 2352-152X. DOI: https://doi.org/10.1016/j.est.
 2021.103425. URL: https://www.sciencedirect.com/science/article/pii/ S2352152X21011117.
- John Cannarella and Craig B. Arnold. "State of health and charge measurements in lithium-ion batteries using mechanical stress". In: Journal of Power Sources 269 (2014), pp. 7–14. ISSN: 0378-7753. DOI: https://doi.org/10.1016/j.jpowsour.
 2014.07.003. URL: https://www.sciencedirect.com/science/article/pii/ S0378775314010453.
- ³⁹⁶ [11] AJ Louli, LD Ellis, and JR Dahn. "Operando pressure measurements reveal solid
 ³⁹⁷ electrolyte interphase growth to rank Li-ion cell performance". In: *Joule* 3.3 (2019),
 ³⁹⁸ pp. 745–761.
- Emanuele Michelini et al. "Experimental Investigation on Reversible Swelling Mechanisms of Lithium-Ion Batteries under a Varying Preload Force". In: *Batteries* 9.4 (2023). ISSN: 2313-0105. DOI: 10.3390/batteries9040218. URL: https://www.mdpi.com/2313-0105/9/4/218.
- ⁴⁰³ [13] Jean-Marie Doux et al. "Pressure effects on sulfide electrolytes for all solid-state
 ⁴⁰⁴ batteries". In: J. Mater. Chem. A 8 (10 2020), pp. 5049–5055. DOI: 10.1039/
 ⁴⁰⁵ C9TA12889A.
- ⁴⁰⁶ [14] Wesley Chang et al. "Evolving contact mechanics and microstructure formation dy⁴⁰⁷ namics of the lithium metal-Li7La3Zr2O12 interface". In: *Nature Communications*⁴⁰⁸ 12 (Nov. 2021), p. 6369. DOI: 10.1038/s41467-021-26632-x.
- [15] Xin Zhang et al. "Pressure-Driven Interface Evolution in Solid-State Lithium Metal
 Batteries". In: Cell Reports Physical Science 1.2 (2020), p. 100012. ISSN: 26663864. DOI: https://doi.org/10.1016/j.xcrp.2019.100012. URL: https:
 //www.sciencedirect.com/science/article/pii/S266638641930013X.
- [16] Alexander J Louli et al. "Exploring the impact of mechanical pressure on the performance of anode-free lithium metal cells". In: *Journal of The Electrochemical Society* 166.8 (2019), A1291–A1299.
- [17] Cameron Martin et al. "Cycling Lithium Metal on Graphite to Form Hybrid LithiumIon/Lithium Metal Cells". In: *Joule* 4 (2020), pp. 1296–1310. DOI: https://doi.
 org/10.1016/j.joule.2020.04.003.
- ⁴¹⁹ [18] Gregory L Plett. *Battery management systems. Volume I, Battery modeling.* Tech-⁴²⁰ nology & Engineering, 2015.
- [19] Gert Berckmans et al. "Electrical Characterization and Micro X-ray Computed To mography Analysis of Next-Generation Silicon Alloy Lithium-Ion Cells". In: World
 Electric Vehicle Journal 9.3 (2018). ISSN: 2032-6653. DOI: 10.3390/wevj9030043.
 URL: https://www.mdpi.com/2032-6653/9/3/43.

- ⁴²⁵ [20] Charles Monroe and John Newman. "The Effect of Interfacial Deformation on Electrodeposition Kinetics". In: *Journal of The Electrochemical Society* 151 (June 2004),
 ⁴²⁷ A880–A886. DOI: 10.1149/1.1710893.
- [21] Sangil Hyun and Mark O. Robbins. "Elastic contact between rough surfaces: Effect
 of roughness at large and small wavelengths". In: *Tribology International* 40.10
 (2007). Tribology at the Interface: Proceedings of the 33rd Leeds-Lyon Symposium
 on Tribology (Leeds, 2006), pp. 1413–1422. ISSN: 0301-679X. DOI: https://doi.
 org/10.1016/j.triboint.2007.02.003. URL: https://www.sciencedirect.
 com/science/article/pii/S0301679X07000369.
- Ioachim Larsson, Shiro Biwa, and Bertil Storåkers. "Inelastic flattening of rough surfaces". In: *Mechanics of Materials* 31.1 (1999), pp. 29–41. ISSN: 0167-6636. DOI: https://doi.org/10.1016/S0167-6636(98)00046-5. URL: https://www.
 sciencedirect.com/science/article/pii/S0167663698000465.
- H. M. Stanley and T. Kato. "An FFT-Based Method for Rough Surface Contact". In: *Journal of Tribology* 119.3 (July 1997), pp. 481–485. ISSN: 0742-4787.
 DOI: 10.1115/1.2833523. eprint: https://asmedigitalcollection.asme.
 org/tribology/article-pdf/119/3/481/5602831/481_1.pdf. URL: https:
 //doi.org/10.1115/1.2833523.
- [24] Shashank Arora, Weixiang Shen, and Ajay Kapoor. "Review of mechanical design and strategic placement technique of a robust battery pack for electric vehicles". In: *Renewable and Sustainable Energy Reviews* 60 (2016), pp. 1319–1331. ISSN: 1364-0321. DOI: https://doi.org/10.1016/j.rser.2016.03.013. URL: https: //www.sciencedirect.com/science/article/pii/S1364032116002483.
- Lip Huat Saw, Yonghuang Ye, and Andrew A.O. Tay. "Integration issues of lithiumion battery into electric vehicles battery pack". In: Journal of Cleaner Production
 113 (2016), pp. 1032–1045. ISSN: 0959-6526. DOI: https://doi.org/10.1016/
 j.jclepro.2015.11.011. URL: https://www.sciencedirect.com/science/
 article/pii/S0959652615016406.
- Gert Berckmans et al. "Analysis of the effect of applying external mechanical pressure on next generation silicon alloy lithium-ion cells". In: *Electrochimica Acta*306 (2019), pp. 387–395. ISSN: 0013-4686. DOI: https://doi.org/10.1016/j.
 electacta.2019.03.138. URL: https://www.sciencedirect.com/science/
 article/pii/S0013468619305614.
- Lysander De Sutter et al. "Mechanical behavior of Silicon-Graphite pouch cells under external compressive load: Implications and opportunities for battery pack design". In: *Journal of Power Sources* 451 (2020), p. 227774. ISSN: 0378-7753. DOI: https://doi.org/10.1016/j.jpowsour.2020.227774. URL: https://www.
 sciencedirect.com/science/article/pii/S037877532030077X.

- ⁴⁶³ [28] Bernhard Bitzer and Andreas Gruhle. "A new method for detecting lithium plat⁴⁶⁴ ing by measuring the cell thickness". In: *Journal of Power Sources* 262 (2014),
 ⁴⁶⁵ pp. 297-302. ISSN: 0378-7753. DOI: https://doi.org/10.1016/j.jpowsour.
 ⁴⁶⁶ 2014.03.142. URL: https://www.sciencedirect.com/science/article/pii/
 ⁴⁶⁷ S0378775314004753.
- ⁴⁶⁸ [29] Davide Clerici, Francesco Mocera, and Aurelio Somà. "Electrochemical-mechanical multi-scale model and validation with thickness change measurements in prismatic
 ⁴⁷⁰ lithium-ion batteries". In: *Journal of Power Sources* 542 (2022), p. 231735. ISSN:
 ⁴⁷¹ 0378-7753. DOI: https://doi.org/10.1016/j.jpowsour.2022.231735. URL:
 ⁴⁷² https://www.sciencedirect.com/science/article/pii/S0378775322007297.
- [30] B. Rieger et al. "Multi-scale investigation of thickness changes in a commercial pouch type lithium-ion battery". In: *Journal of Energy Storage* 6 (2016), pp. 213–221. ISSN: 2352-152X. DOI: https://doi.org/10.1016/j.est.2016.01.006. URL:
 https://www.sciencedirect.com/science/article/pii/S2352152X16300068.

477 [31] Katie Lukow. *MBPF*. 2022. DOI: https://doi.org/10.5281/zenodo.7509368.
 478 URL: https://github.com/katielukow/MBPF.

- High Voltage Energy Storage Group. Battery Testing Consortium Protocol. 2022.
 URL: https://github.com/HVES-Battery-Testing-Consortium/LG-HG2.
- ⁴⁸¹ [33] V.G. Choudhari, Dr A.S. Dhoble, and T.M. Sathe. "A review on effect of heat generation and various thermal management systems for lithium ion battery used for electric vehicle". In: *Journal of Energy Storage* 32 (2020), p. 101729. ISSN:
 ⁴⁸⁴ 2352-152X. DOI: https://doi.org/10.1016/j.est.2020.101729. URL: https:
 ⁴⁸⁵ //www.sciencedirect.com/science/article/pii/S2352152X20315668.
- Yongqi Xie et al. "Experimental and analytical study on heat generation characteristics of a lithium-ion power battery". In: International Journal of Heat and Mass Transfer 122 (2018), pp. 884–894. ISSN: 0017-9310. DOI: https://doi.org/10.
 1016/j.ijheatmasstransfer.2018.02.038. URL: https://www.sciencedirect.
 com/science/article/pii/S001793101733421X.
- Ki-Yong Oh and Bogdan I. Epureanu. "A novel thermal swelling model for a rechargeable lithium-ion battery cell". In: *Journal of Power Sources* 303 (2016),
 pp. 86-96. ISSN: 0378-7753. DOI: https://doi.org/10.1016/j.jpowsour.
 2015.10.085. URL: https://www.sciencedirect.com/science/article/pii/
 S0378775315304730.
- ⁴⁹⁶ [36] Yan Zhao et al. "Localized swelling inhomogeneity detection in lithium ion cells
 ⁴⁹⁷ using multi-dimensional laser scanning". In: *Journal of The Electrochemical Society*⁴⁹⁸ 166.2 (2019), A27.
- ⁴⁹⁹ [37] Anup Barai et al. "A study of the influence of measurement timescale on internal
 ⁵⁰⁰ resistance characterisation methodologies for lithium-ion cells". In: *Scientific reports*⁵⁰¹ 8.1 (2018), pp. 1–13.

[38] Wladislaw Waag, Stefan Käbitz, and Dirk Uwe Sauer. "Experimental investigation of the lithium-ion battery impedance characteristic at various conditions and aging states and its influence on the application". In: *Applied Energy* 102 (2013).
Special Issue on Advances in sustainable biofuel production and use - XIX International Symposium on Alcohol Fuels - ISAF, pp. 885–897. ISSN: 0306-2619.
DOI: https://doi.org/10.1016/j.apenergy.2012.09.030. URL: https: //www.sciencedirect.com/science/article/pii/S030626191200671X.

⁵⁰⁹ [39] Tobias Teufl et al. "State of charge dependent resistance build-up in Li-and Mn⁵¹⁰ rich layered oxides during lithium extraction and insertion". In: *Journal of The*⁵¹¹ *Electrochemical Society* 166.6 (2019), A1275.

Jianming Zheng et al. "Electrochemical kinetics and performance of layered composite cathode material Li [Li0. 2Ni0. 2Mn0. 6] O2". In: Journal of The Electrochemical
Society 160.11 (2013), A2212.

[41] Sanketh R Gowda et al. "Examining the electrochemical impedance at low states of charge in lithium-and manganese-rich layered transition-metal oxide electrodes".
In: Journal of The Electrochemical Society 162.7 (2015), A1374.

518 Data Availability

For questions in regard to obtaining test data, please contact the High Voltage Energy Storage group at Oxford Brookes University: https://hves.brookes.ac.uk.